Model Electrode Studies of the Electrostatic Interaction between Electrochemically Dissolved Pt Ions and RuO₂ Nanosheets

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Platinum supported on carbon (Pt/C) is widely used as a cathode catalyst in polymer electrolyte fuel cells because of its high oxygen reduction reaction (ORR) activity. The loss of electrocatalytic activity during fuel cell operation is a detrimental factor to the useful lifetime of commercial polymer electrolyte fuel cell systems. Hence, there is a strong demand to improve the durability of electrocatalyst to realize the wide-spread commercialization of polymer electrolyte fuel cells. Numerous studies have clarified that dissolution, migration and/or sintering of platinum nanoparticles on carbon are vital degradation factors of the cathode catalyst.1–4 Oxides that are stable under acidic and oxidizing conditions have been suggested to enhance the durability of Pt as cathode catalysts. For example, SnO₂ has been proposed as an alternative support to replace carbon to enhance the durability of Pt catalyst due to its resistance to corrosion.5 TiO₂ added to Pt/C was suggested to anchor platinum particles, preventing agglomeration and coalescence during durability testing.6 Carbon supported Pt covered with a thin layer of SiO₂ has been shown to exhibit high stability during potential cycling in H₂SO₄ electrolyte.7,8 The foundation of the increase in durability due to the addition of these oxides is not well understood. In addition, due to the poor conductivity of these oxides, the original properties of Pt/C are often inevitably obstructed, which includes the loss of initial electrochemical surface area (ECSA) and ORR activity with the addition of oxides.

Contrary to most other oxide systems, RuO₂ nanostructures possess excellent electronic conductivity and electrochemical stability, making them an ideal additive that would not obstruct electrode kinetics. Indeed, we and others have found that the combination of RuO₂ nanostructures and Pt nanoparticles can enhance ORR properties. For example, incorporation of carbon-supported RuO₂ (RuO₂/C) to Pt was found to ameliorate both stability and activity of the cathode catalyst.9 We have reported that the ORR activity and durability of commercial Pt/C are improved by the modification with RuO₂ nanosheets.10–12 Because of the complex structure of the porous carbon support in commercial catalyst coupled with the ultimate thickness of the nanosheet of 1 nm, it was difficult to obtain conclusive evidence to explain the mechanism of the improvement on the catalyst durability. Therefore, studies of a simple model electrode system are desired to understand the enhanced durability of Pt/C electrocatalyst modified with RuO₂ nanosheets. The requirement for an ideal model electrode is that the electrode should be accessible to certain analytical techniques before, during and after the involved catalytic process. Model electrodes are generally prepared by depositing platinum nanoparticles on two-dimensional, planar carbon substrate such as glassy carbon (GC) or highly oriented pyrolytic graphite (HOPG) by electrodeposition, vapor deposition or lithography.13–20 Surface analysis methods, such as scanning tunneling microscopy (STM), atomic force microscopy (AFM), scanning electron microscopy (SEM) are often used to characterize the electrochemical process of model electrodes.15–20

In this work, model electrode studies with HOPG substrate were performed to investigate the interaction between RuO₂ nanosheets and dissolved Pt ions which were generated electrochemically in an attempt to clarify the enhanced durability of Pt/C electrocatalyst modified with the nanosheets. STM and AFM were utilized to observe the adsorption and reduction behavior of dissolved Pt ions on the RuO₂ nanosheet/HOPG model electrode.

**Experimental**

Ruthenium oxide nanosheets were synthesized by elemental exfoliation of an ion-exchangeable layered potassium ruthenate (K₂RuO₂.₁·nH₂O).²¹,²² Proton-exchange of the interlayer potassium was conducted with 1 mol dm⁻³ HCl for 3 days at 60 °C, resulting in the layered ruthenic acid (H₃RuO₂.₁·0.9H₂O). The layered ruthenic acid was added to a tetrabutylammonium hydroxide (TBAOH) aqueous solution with the molar ratio of TBA ions to the exchangeable protons in H₃RuO₂.₁·nH₂O adjusted to TBA⁺/H⁺ = 1.5. The dispersion was vigorously shaken for 10 days to exfoliate the layered ruthenate into elementary RuO₂ nanosheets. Non-exfoliated impurity was removed by centrifugation at 2000 rpm for 30 min. The as-exfoliated nanosheet colloidal was finally diluted to 0.1 (g-RuO₂)L⁻¹ with ultrapure water (Milli-Q, > 18 MΩ cm). HOPG (Bruker, ZYH-grade, 12 × 12 mm²) was freshly cleaved using adhesive tape and then immersed into the nanosheet colloidal for 2 minutes to coat RuO₂ nanosheets. The nanosheets coated HOPG (RuO₂ nanosheet/HOPG) was rinsed with ultrapure water and then dried under vacuum.

Dissolved Pt ions were generated electrochemically by potential cycling a Pt mesh working electrode 500 times between 0.05 to 1.4 V at a scan rate of 100 mV s⁻¹ in a three-electrode electrochemical cell in O₂ saturated 0.5 M H₂SO₄ (60 °C). RuO₂ nanosheet/HOPG was immersed in this solution for 10 minutes at room temperature to adsorb the dissolved Pt ions.

STM measurements were performed using a scanning tunneling microscope (STM, Bruker, Digital Instruments Nanoscope III D ADC 5) equipped with a 10 μm scanner (HD-8I, 2399DI) with the set point current of 1 nA. In order to avoid the disturbance of Faraday current on tunneling current, the tip of the PtIr probe (Bruker, Pt-ECM10, 14 mm length) was partially coated by nail polish as an insulating layer. The geometric area of the working electrode (WE) in contact with the...
electrolyte was 0.5 cm$^2$. Images were recorded using SiN probe (Bruker, SNL-10). The test sample acted as the working electrode and two high purity Pt wires with a diameter of 0.1 mm acted as the reference electrode (RE) and the counter electrode (CE). An electrochemical-atomic force microscope (EC-AFM, Bruker, Digital Instruments Nanoscope III D ADC 5) equipped with a bi-potentiostat was used. Electrochemical reduction of adsorbed Pt ions was conducted by slow-scan linear sweep voltammetry from 1.1 to 0 V (cathodic scan) at 0.25 mV s$^{-1}$. The time for the image acquisition was approximately 5 minutes. Both STM and AFM images were processed using Nanoscope V531r1 software.

The ORR activity of reduced Pt ions adsorbed on RuO$_2$ nanosheets was investigated in a standard three-electrode electrochemical cell with 0.5 M H$_2$SO$_4$ as electrolyte. Carbon fiber (Toho, Tenax-J, HTS40 E13) was used as the counter electrode and a reversible hydrogen electrode (RHE) was used as the reference electrode. Electrochemical reduction was conducted by slow-scan linear sweep voltammetry from 1.1 to 0 V in O$_2$-saturated 0.5 M H$_2$SO$_4$ (25 $^\circ$C). Linear sweep voltammetry (LSV) was then conducted at 10 mV s$^{-1}$ from 0.05 to 1.2 V vs. RHE in O$_2$-saturated 0.5 M H$_2$SO$_4$. The LSV data collected in N$_2$-purged electrolyte was used as background and subtracted from the data recorded in O$_2$-saturated 0.5 M H$_2$SO$_4$.

**Results and Discussion**

A typical STM image of RuO$_2$ nanosheets supported on HOPG observed in 0.5 M H$_2$SO$_4$ is shown in Figure 1. The height profiles of a number of specimens showed that the nanosheets have a thickness of 1 $\pm$ 0.1 nm with several hundreds of nanometers in lateral size, revealing complete exfoliation into monolayer. Respective STM images in air were also collected and were in good agreement to our previous AFM data.$^{22,23}$ The thickness of the RuO$_2$ nanosheets both in air and in 0.5 M H$_2$SO$_4$ were the same. This indicates that the RuO$_2$ nanosheets are fairly well fixed on the HOPG substrate and does not drift away from the substrate in aqueous solution. This allows to investigate the adsorption of dissolved Pt ions on the surface of RuO$_2$ nanosheets using STM or AFM.

Spontaneous formation of Pt nanoparticles on as-cleaved HOPG surface in chloride electrolytes containing PtCl$_6^{2-}$ is known to occur.$^{13,24}$ The driving force for this process has been suggested to be related to the presence of incompletely oxidized functionalities existing at terraces and kink sites on freshly cleaved HOPG surface.$^{13}$ Here, we immersed an as-cleaved HOPG into 0.5 M H$_2$SO$_4$ containing electrochemically dissolved Pt ions to evaluate the chemical interaction between dissolved Pt ions and HOPG. As seen in Figure 2, individual Pt nanoparticles can be observed at the edge sites of the HOPG surface, which indicates that dissolved Pt ions have properties similar to PtCl$_6^{2-}$. The individual Pt nanoparticles have an average diameter of 10–20 nm with height of about 1.5–2.5 nm, similar in geometry to Pt nanoparticles reduced on the edge sites of HOPG from PtCl$_6^{2-}$.$^{13,24}$ It is assumed that the dissolved Pt ions are spontaneously reduced by the incompletely oxidized functionalities at the edge sites, analogous to the case for PtCl$_6^{2-}$.$^{13}$ The characteristic plate-like morphology may come from the high mobility of as-reduced Pt on the HOPG surface leading to a two-dimensional growth.

In order to investigate the interaction between RuO$_2$ nanosheets and dissolved Pt ions, the RuO$_2$ nanosheet/HOPG was immersed into 0.5 M H$_2$SO$_4$ containing dissolved Pt ions, dried, and characterized by STM. Figure 3a shows a 3D topographic STM image of RuO$_2$ nanosheet/HOPG after adsorption of dissolved Pt ions, showing island-like deposits on the surface of nanosheets. Figure 3b shows the top-view STM image corresponding to Figure 3a. The deposits are aligned parallel to each other, possibly reflecting the atomic arrangement of the the nanosheet surface. Note that no deposits appear on the surface of HOPG. The height profile (Figure 3c) shows that the island-like deposits are several tens of nanometers in diameter and $\sim$1.8 nm in height.

Dissolved Pt ions in sulfuric acid have been detected as cationic Pt$^{2+}$ and Pt$^{4+}$ species.$^{25}$ The solvation shell of hydrated Pt ion was...
calculated as eight water molecules, making the diameter of the hydrated Pt ion to be approximately 0.79 nm. A dense monolayer of hydrated Pt ion should then give a coverage of $1.85 \times 10^6$ Pt$^{n+} \mu\text{m}^{-2}$. As the average height of the island-like deposits is $\sim1.8$ nm (Figure 3), the deposits should be formed by 2 or 3 layers of hydrated Pt ion. The coverage of the deposits was 22%, which translates to a coverage of 50% for a monolayer adsorption. The amount of hydrated Pt ion on the surface of nanosheet is thus calculated to be $0.93 \times 10^6$ Pt$^{n+} \mu\text{m}^{-2}$.

Next, we consider the amount of Pt$^{n+}$ adsorption theoretically possible based on the surface charge of RuO$_2$ nanosheet. Assuming that the crystal structure of RuO$_2$ nanosheet is close to rutile-type RuO$_2$ (density of 6.97 g cm$^{-3}$), the amount of Ru atoms in a 1 nm thick sheet is calculated as $3.07 \times 10^7$ atoms $\mu\text{m}^{-2}$. RuO$_2$ nanosheet has a formal composition of $(\text{RuO}_2)_{x \geq 2}$, which means that the number of negative charge will be $6.14 \times 10^8$ sites $\mu\text{m}^{-2}$. Considering that only one side of nanosheet is exposed to the electrolyte, the actual number of negative charge should be half, thus giving $3.07 \times 10^8$ sites $\mu\text{m}^{-2}$. Then, the saturated adsorption amount for Pt$^{2+}$ and Pt$^{4+}$ should be $1.54 \times 10^6$ and $0.77 \times 10^6$ atoms $\mu\text{m}^{-2}$, respectively. Since the ratio of Pt$^{2+}$ to Pt$^{4+}$ has been detected as approximately 1.3 in H$_2$SO$_4$, the final amount of adsorbed Pt ions should be $0.96 \times 10^6$ atoms $\mu\text{m}^{-2}$. Thus, the amount of hydrated Pt ion adsorbed on nanosheet estimated from the STM images (Figure 3) shows a close match to the estimated saturated adsorption amount of cationic Pt ions on RuO$_2$ nanosheet surface. This undoubtedly implies that the adsorption of Pt$^{n+}$ on the surface of RuO$_2$ nanosheet is due to the strong electrostatic interactions between hydrated Pt cations and negatively charged RuO$_2$ nanosheets.

The adsorbed Pt ions on RuO$_2$ nanosheet were then electrochemically reduced by performing a cathodic scan from 1.1 to 0 V at a slow scan rate of 0.25 mV s$^{-1}$. Figure 4 shows the tapping-mode AFM image after electrochemical reduction. Pt nanoparticles with average height of 2.0 ± 0.7 nm and width/length ranging from 2 to 20 nm were observed on the surface of RuO$_2$ nanosheet. The ORR activity of the Pt nanoparticles decorated on RuO$_2$ nanosheets was investigated in a standard three-electrode electrochemical cell. Figure 5 shows the linear sweep voltammogram of RuO$_2$ nanosheet/HOPG with adsorbed Pt ions after electrochemical reduction. It is clear that Pt nanoparticles supported on the nanosheet show ORR activity. We consider that the ORR activity shown by the reduced Pt nanoparticles on nanosheet contributes to the higher ORR activity retention rate in practical catalyst. As discussed in our previous papers, RuO$_2$ nanosheet modified Pt/C electrocatalyst shows enhanced durability compared to non-modified Pt/C. Certainly, the degradation rate of carbon blacks used in practical catalysts and HOPG used in this model electrode study should be quite different due to the difference in degree of graphitization. Nonetheless, this model electrode study reveals that there is a strong electrostatic interaction between the additive RuO$_2$ nanosheets and dissolved Pt ions. This model electrode study clarifies that the enhancement in durability by addition of RuO$_2$ nanosheet to practical Pt/C catalyst can be attributed, at least in part, to the negatively charged RuO$_2$ nanosheets acting as trapping sites to mitigate the migration of adsorbed Pt ions into the electrolyte.
This value closely matched the full saturated adsorption of Pt ions on RuO₂ nanosheets, thereby impede loss of Pt during potential cycling in practical supported catalysts.

Conclusions

A RuO₂ nanosheet/HOPG model electrode was prepared by dipping HOPG into an aqueous solution of RuO₂ nanosheet. The adsorption of electrochemically generated dissolved Pt ions in sulfuric acid on RuO₂ nanosheet/HOPG was detected by STM. Based on the geometries of the deposits, the amount of hydrated Pt ion on the surface of nanosheets was estimated to be 0.93 × 10⁶ Pt⁺⁺ μm⁻², which value closely matched the full saturated adsorption of Pt ions of 0.96 × 10⁶ Pt⁺⁺ μm⁻² estimated based on the negative charge of the nanosheets. Thus, we conclude that the driving force for adsorption of Pt ions on RuO₂ nanosheets is the strong electrostatic interaction between positively charged Pt ions and negatively charged RuO₂ nanosheets. The adsorbed Pt ions could easily be reduced to Pt nanoparticles by electrochemical reduction, which exhibited ORR activity. The electrostatic interaction between Pt ions and RuO₂ nanosheets should facilitate trapping and re-deposition of the Pt ions on RuO₂ nanosheets, thereby impede loss of Pt during potential cycling in practical supported catalysts.

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References


Figure 5. LSV of RuO₂ nanosheet/HOPG with adsorbed Pt ions after electrochemical reduction (from 0.05 to 1.2 V vs. RHE, 10 mV s⁻¹, 25°C).